Imaging Bulk and Edge Transport near the Dirac Point in Graphene Moiré Superlattices

Ziwei Dou,*,†‡ Sei Morikawa,‡ Alessandro Cresti,§ Shu-Wei Wang,† Charles G. Smith,‡ Christos Melios,‖ Olga Kazakova,‖ Kenji Watanabe,‡ Takashi Taniguchi,* Satoru Masubuchi,‖ Tomoki Machida,‖ and Malcolm R. Connolly*§∥

* Cavendish Laboratory, Department of Physics, University of Cambridge, Cambridge CB3 0HE, United Kingdom
† Institute of Industrial Science, University of Tokyo, 4-6-1 Komaba, Meguro, Tokyo 153-8505, Japan
‡ Univ. Grenoble Alpes, CNRS, Grenoble INP, IMEP-LaHC, F-38000 Grenoble, France
§ National Physical Laboratory, Hampton Road, Teddington TW11 0LW, United Kingdom
‖ Institute for Nano Quantum Information Electronics, University of Tokyo, 4-6-1 Komaba, Meguro, Tokyo 153-8505, Japan

Supporting Information

ABSTRACT: Van der Waals structures formed by aligning monolayer graphene with insulating layers of hexagonal boron nitride exhibit a moiré superlattice that is expected to break sublattice symmetry. Despite an energy gap of several tens of millielectronvolts opening in the Dirac spectrum, electrical resistivity remains lower than expected at low temperature and varies between devices. While subgap states are likely to play a role in this behavior, their precise nature is unclear. We present a scanning gate microscopy study of moiré superlattice devices with comparable activation energy but with different charge disorder levels. In the device with higher charge impurity (∼10⁻⁴ cm⁻²) and lower resistivity (∼10 kΩ) at the Dirac point we observe current flow along the graphene edges. Combined with simulations, our measurements suggest that enhanced edge doping is responsible for this effect. In addition, a device with low charge impurity (∼10⁹ cm⁻²) and higher resistivity (∼100 kΩ) shows subgap states in the bulk, consistent with the absence of shunting by edge currents.

KEYWORDS: Graphene moiré superlattice, gapped Dirac Fermion system, scanning gate microscopy, topological edge states, valley Hall effects

The mobility of charge carriers in two-dimensional (2D) crystals of graphene is enhanced by encapsulating them between atomically flat layers of hexagonal boron-nitride (hBN). Alongside improvements in device quality has come a rich array of physics arising from the van der Waals interaction between the 2D layers. Moiré superlattices,²⁻⁵ for instance, are formed by aligning the graphene lattice within a few degrees of the hBN [Figure 1a]. Strain-relaxation⁴,⁵ instance, are formed by aligning the graphene lattice within a λ

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symmetry. Understanding the current distribution in gapped graphene devices near the CNP is therefore a high concern. Indeed, recent Josephson interferometry in very short (∼100 nm) graphene junctions between superconducting Nb contacts shows a transition from bulk-to-edge transport as the energy approaches the CNP in both gapped BLG and MLG.

In this work, we use scanning gate microscopy (SGM) to examine micron-sized MLG moiré superlattices. In particular, we focus on the current distribution near the main DP, where the gap due to moiré-induced broken inversion-symmetry and the mechanism for finite subgap conduction at low temperature is still not well understood. We present data taken from two devices with comparable activation energies ($E_g/2 \sim 14$ meV) but different levels of disorder-induced energy broadening, $E_g$. In device D1, $E_g > E_g/2$, and the resistivity at the CNP saturates at ∼10 kΩ, comparable to the $h/e^2$, whereas in device D2, $E_g < E_g/2$, and the CNP resistivity (∼100 kΩ) is much higher than the $h/e^2$. Our SGM images show a response at the edges in D1 near the CNP. Despite its lower disorder, the SGM response in D2 is in the bulk. Using tight-binding simulations as a guide, we are able to explain our results by enhanced doping at the edges of D1. Our results therefore suggest that whereas edge transport plays an important role in shunting the gapped bulk in MLG moiré superlattice devices, it probably originates from external factors such as disorder and inhomogeneous doping rather than universal properties of gapped Dirac spectrum.

Our MLG encapsulated moiré superlattice devices [Figure 1a] are fabricated by mechanical exfoliation and the “pick-up” transfer technique of atomic layers. The MLG is sandwiched between a 50 nm top layer hBN and a 70 nm bottom layer hBN, and the whole structure is placed on a ∼290 nm SiO$_2$ formed on a doped Si substrate used for applying a back-gate voltage ($V_{BG}$). The devices are then patterned into Hall bars and Cr/Pt/Au ohmic contacts are made via one-dimensional edge contacts. An optical image of the device and circuit schematic is shown in Figure 1a. We use standard low-frequency lock-in techniques to determine the four-terminal resistance $R_{4T} = V_{xx}/I_{SD}$. For D1, we first confirm the presence of a moiré superlattice by observing secondary Dirac points (SDPs) in $R_{4T}(n)$, a well-known signature resulting from the modified bandstructure of moiré MLG [Figure 1b]. The carrier density $n$ is calculated by $C_{BG}(V_{BG} - V_{CNP})$ where $C_{BG} = 1.124 \times 10^{-4}$ F/m$^2$ is the capacitance per area to the backgate extracted from quantum Hall measurements, and $V_{CNP} = 1.25$ V is the back-gate voltage at the charge neutrality point. From the SDP carrier densities we estimate $\lambda_M$ of ∼11 nm.

Figure 1c shows the resistivity $\rho$ (calculated from $R_{4T} \sim \rho L/W$, where $L = 11.5$ µm and $W = 1.3$ µm are the length and width, respectively) measured at $T = 4$ K as a function of $n$. From the full width at half-maximum (fwhm) of the Dirac peak (∼4 × $10^{-2} \text{cm}^{-2}$), we estimate disorder-induced broadening of ∼24 meV.$^{3,4,6}$ Figure 1d shows the temperature dependence of $\rho$ in the range of 4–200 K. The maximum resistivity at the CNP ($\rho_{CNP}$) is extracted from back-gate sweeps performed at each temperature and plotted against inverse temperature, resulting in an Arrhenius plot. We identify the following three regimes. (I) at high temperature (>70 K) the data are well described by a simple model for thermally activated transport across an energy gap $E_g$, $\rho_{CNP} \sim \exp\left(\frac{E_g}{2kT}\right)$, where $k$ is the Boltzmann constant. By fitting the data to this expression (red dashed line), we extract $E_g/2 \sim 160$ K (∼14 meV), consistent with previous work$^{3,4,6}$ and the gap expected from the estimated $\lambda_M$. (II) At intermediate temperatures (gray hatched lines), the weak temperature dependence is consistent with variable range hopping, and by fitting the resistivity to $\rho \sim \exp\left((T_0/T)^{1/2}\right)$, we deduce $T_0 \sim 2$ K.$^{5,28}$ (III) At low temperature, the resistivity saturates at ∼10 kΩ, comparable to the resistance quantum $h/e^2$ and consistent with earlier works.$^{3,4}$

In an effort to resolve the microscopic mechanism behind the flattening of $\rho_{CNP}(T)$ in regime (III), we use SGM at $T \sim 4$ K. SGM is a well-established technique for visualizing local variations in the electronic properties of 2DESs and involves monitoring the conductance of a mesoscopic device.
while scanning a sharp metallic tip over its surface. Figure 2a illustrates the SGM setup used in our experiments, where a constant current ($I_{SD} = 100$ nA) is driven between the two ends of the superlattice while measuring the voltage $V_{xx}$. The tip is lifted $\sim 100$ nm above the top hBN layer and locally gates the graphene. The scanned area in (b,c) are marked by the red dashed rectangle (tip not to scale, scale bar: 1 $\mu$m). SGM image of $R_{4T} = V_{xx}/I_{SD}$ as a function of tip position when globally (b) $n \sim -2.5 \times 10^{10}$ cm$^{-2}$ and (c) $n \sim 0$ cm$^{-2}$. Edges of the device are outlined by dotted lines and each grid represents 1 $\mu$m. Blue dashed oval depicts the region to the long-range tip gating effect. Insets show $R_{4T}(y)$ averaged over all $x$. (d) Sequence of higher-resolution SGM images of $R_{4T}$ in the area marked by the black rectangle in (b). The images are taken at different $n$: $-2.11 \times 10^{10}$ cm$^{-2}$ (blue); $-1.05 \times 10^{10}$ cm$^{-2}$ to $-2.11 \times 10^{10}$ cm$^{-2}$ intervals (green to magenta); $0.14 \times 10^{10}$ to $0.70 \times 10^{10}$ cm$^{-2}$ with $-0.14 \times 10^{10}$ cm$^{-2}$ intervals (magenta to gray). The device edges are outlined with dotted lines; each grid represents 0.5 $\mu$m. Evolution of a “hotspots” is marked by blue dashed circle. (e) SGS as a function of $n$ along the black arrow in (d). The SGS is plotted in five panels with different color scales for clarity. Corresponding values of $n$ for the images in (d) are marked with colored circles. (f) Same SGS as (e) but with $V_T^* \sim +1.5$ V. (g) Reference $R_{4T}(n)$ without the tip. Plots in panels e–g share the same horizontal axis. The CNP is marked by the vertical dashed line.
the effective tip potential $V_{T^*} = V_T - V_0 = -0.5$ V, reducing the resistance of the hole-doped channel and producing the feature within the oval dashed line. When the tip is over the channel, $V_0$ is measured to be $\sim -1$ V, and $V_{T^*} = 0.5$ V, giving rise to the observed resistance enhancement (see Supporting Information, Section VI). With the global doping level at the CNP [Figure 2c, bottom image], a $\sim 1$ kΩ enhancement in $R_{TT}$ is seen only when the tip is over the edges of the flake [see plot of averaged $R_{TT}$ (y) in Figure 2c], which cannot be explained by a spatially dependent contact potential (see Supporting Information, Section VI).

We investigate more closely by capturing a sequence of high-resolution scans at different carrier densities over the range from $-2.1 \times 10^{10}$ cm$^{-2}$ to $0.7 \times 10^{10}$ cm$^{-2}$ (note the unequal carrier density intervals denoted in the caption) and plot the SGM micrographs in Figures 2d. The scanned region is outlined in Figure 2b by the black rectangular box and the device edges are marked as black dotted lines. The “hotspots” (defined here as an isolated region of tip position where the resistance is higher than the background) that are initially over the bulk channel split laterally and migrate toward the edges. At the CNP, some hotspots fragment further along the edge (see the dashed circles). To track the bulk to edge transition we perform scanning gate spectroscopy (SGS), which involves stepping the tip position along a line and sweeping the global carrier density at each point. Plotting the resulting $R_{TT}(n)$ allows us to determine the evolution of a particular feature with both high energy and high spatial resolution in one dimension.
altogether with more positive tip bias as the device becomes electron doped, such features disappear. Features resembling the transition back to bulk response is seen predominantly occurs for hole doping. While in Figure 2ea about the Dirac point, and the bulk-edge transition CNP, similar to the hotspot features. The data are asymmetric continuously opens until it reaches the device edge at the resistance only in the bulk. Such bulk response the previous micrographs, away from the CNP the tip increases the background resistance with decreasing n. As expected from the previous micrographs, away from the CNP the tip increases the resistance only in the bulk. Such bulk response continuously opens until it reaches the device edge at the CNP, similar to the hotspot features. The data are asymmetric about the Dirac point, and the bulk-edge transition predominantly occurs for hole doping. While in Figure 2e a feature resembling the transition back to bulk response is seen as the device becomes electron doped, such features disappear altogether with more positive tip bias \( V_{\text{tip}}^* \sim +1.5 \text{ V} \) in Figures 2f, where the bulk-edge transition occurs for hole doping only.

Before discussing the edge response in more detail, we examine a second device (D2) with similar properties but with ~3 times lower bulk disorder level. Figure 3a shows an optical image of the device along with the two-terminal measurement schematic for SGM. We first check the transport properties of D2 using the same four-terminal measurement as in Figure 1 on D1. Figure 3b–d shows the Dirac peak, Arrhenius plot of the resistivity at CNP, and the SDPs, respectively. From Figure 3b, the charge carrier density disorder of D2 is \(~0.4 \times 10^{10} \text{ cm}^{-2}\) (or ~7 meV), a factor of 3 lower than D1 (plotted in reference as the dashed line) and is comparable with the best quality reported in literature. The temperature dependence [Figure 3c] shows an activated region with similar transport gap ~28 meV and plateau at low temperature. Figure 3d confirms the moiré superlattice of the sample with SDPs translated to similar moiré wavelength ~11 nm. Therefore, D2 has the same gap size as D1 but with lower bulk disorder. From earlier proposals, \(^{10,12,18}\) edge transport is expected to be more pronounced in D2. To test this, we perform the equivalent SGM and SGS using two-terminal measurements [Figure 3a]. Figure 3e shows a series of SGM images of \( R_{\text{SG}} = V_{\text{SD}}/I_{\text{SD}} \) with constant voltage \( V_{\text{SD}} = 1 \text{ mV} \) taken with the same scan conditions as Figure 2 at different values of n around the CNP. We do not observe an edge response but only a few regions in the bulk where the tip can affect transport [see the dashed circles in Figure 3e]. We follow the same method employed for D1 and perform SGS along the y-direction perpendicular to the edge. The spectroscopy plotted in Figure 3f shows a weak gating effect that produces an overall shift in the back-gate voltage of the CNP but with no increase in \( R_{\text{SG}} \) at the edge near the CNP \( R_{\text{SG}}(n) \) without the tip is plotted in Figure 3f for
The strong ~ MΩ variations due to the tip (as large as ~50% of the total $R_{ST}$) are clearly concentrated in the bulk.

We perform SGS by moving the tip along a line, this time parallel to the channel [arrow along x in Figure 3e] and intersecting the sites marked by the dashed circles. The spectroscopy is plotted in Figure 3h. When the tip is in the vicinity of each site, a peak in $R_{ST}(n)$ appears at higher $n$. This peak splits from the main Dirac point and follows a Lorentzian-shaped trajectory as a function of $x$. To emphasize these trajectories we take the derivative $dR_{ST}/dn$ in the area marked by the white box in Figure 3h, and plot the result in Figure 3i. The trajectories can be fitted by Lorentzians with fwhm of ~0.5 μm and origins separated by Δ$x$ ~ 1 μm. Line sections [Figure 3i] show that height of each peak (P2,3) is comparable to the amount by which the main Dirac peak (P1) is reduced (<1 MΩ), suggesting the sites add in series to yield the total resistance at the DP. We confirmed that sites are not associated with inhomogeneous strain in the sample by checking the 2D peak width using scanning Raman spectroscopy (see Supporting Information Section IX). Despite its lower bulk disorder, D2 does not show pronounced edge conduction and the transport appears mediated via subgap states in the bulk. The absence of edge shunting is also noteworthy, given that the resistivity at CNP plateaus are higher than $\mu_e/2$. One natural explanation is that transport is via a series of electron–hole puddles forming a bulk percolation path. High quality devices typically have ~100 nm size puddles, consistent with a small number of conducting sites separated by insulating regions.

Several mechanisms could give rise to the observed edge transport in D1. Possible scenarios include zigzag edge states without or with energy dispersion under more sophisticated Hamiltonians and states with enhanced localization length due to nontrivial bulk band topology from the broken sublattice symmetry. However, because these states are not topologically protected from intervalley scattering, they localize easily and do not participate in transport. The absence of edge response in D2 despite the lower disorder also suggests the resistivity at the DP is governed by extrinsic factors. Two candidates are field-focusing and chemical doping. We rule out the former as it is stronger away from the CNP due to higher screening ability at high carrier density. Moreover, in D1 the CNP is at $V_{BG}$ ~ +1 V. Field focusing should thus induce additional n-type carriers near the edge and a positive applied $V_T$ should decrease the resistance, which is contrary to what we observe experimentally. The most likely mechanism for our observations of D1 is thus chemical doping at the edges. To test this scenario, we perform tight-binding simulations on a staggered sublattice lattice with a staggered sublattice potential ±Δ = ±7.5 meV to break sublattice symmetry and to simulate the moiré-induced gap [Figure 4a]. Short-range random edge roughness is imposed on perfect zigzag edges with (4,1) chirality to simulate localization of the edge states in realistic devices, and a long-ranged bulk disorder is included through randomized Gaussian potential fluctuations with maximum strength of 10 meV whose strength relative to the gap size is in keeping with D1. We use a parabolic profile with 300 nm width and +30 meV amplitude to simulate the enhanced edge potential estimated from the spectroscopic measurements (see Supporting Information Section V). Figure 4b,c shows the local spectral current map $j_0(x,y)$ of the system away from and at the charge neutrality point, respectively. At $E_F = -12.5$ meV [Figure 4b] the bulk is p-type, leading to bulk conduction. At $E_F = 0$ [Figure 4c], bulk transport is gapped out and the current flows at the edges. Note the edge current is present despite the edge roughness. To simulate the SGS experiment, we model the tip by a Lorentzian potential with 200 nm fwhm. We choose a maximum tip-induced potential energy $U_T = -20$ meV, estimated from the effective tip bias $V_T^* = +1.5$ V and a parallel plate capacitor model based on a tip–sample distance ~100 nm. Figure 4d show the simulated SGS with the channel resistance $R_{ST}$ calculated from the transmission coefficient. The SGS [Figure 4d] is plotted in five panels with different color scales similar to Figure 2. The tip passes over the two edges of the channel at $y_T = 0$ and 1 μm ($x_T = 1$ μm) and the energy gap ($\pm7.5$ meV) is marked by the dashed vertical lines. This plot captures some key features of the experimental SGS in Figure 2, namely a clear enhancement of the resistance appears at the edges at around −7.5 meV in the hole-doped regime and diminishes as $E_F$ is tuned away from CNP.

To illustrate how the above SGM features relate to the enhanced doping at the edges, in Figure 4e,f we plot how the underlying current distributions $j_0(x,y)$ in Figure 4b,c change in the presence of the tip, $\Delta j = j(x,y) - j_0(x,y)$. At the edge of the flake, the bandgap (the region between the red and blue solid lines) is lifted to higher energy [Figure 4e, right]. Hence, when $E_F = -12.5$ meV [blue circle in Figure 4d], the edge has a higher hole density than the bulk. Although the current is enhanced along the edges, the effect of the negative tip potential energy is weak because the bulk is conductive. The simulated $\Delta j$ map [Figure 4e] consequently displays weak ambipolar modulations of the current and there is no enhancement at the edges. By contrast, when $E_F = 0$, the $\Delta j$ map [Figure 4f] shows a strong response along the edge because the conduction and valence bands are lowered and the hole density reduces [see band edge profile, Figure 4f]. This gaps out the current flow at the edges and increases the resistance. In the inset of Figure 4f, the same $\Delta j$ map is plotted with positive tip potential. In this case, the edge current is locally enhanced and only weakly affects the global resistance, consistent with the absence of an SGS response. We experimentally observe an edge response only with positively biased tip (or negative tip potential energy), while no edge response is seen with negatively biased tip. This is further consistent with enhanced p-type doping (see Supporting Information Section VII). Within this framework potential disorder along the channel creates the hotspots in Figure 2, while their migration toward the edge reflects the transverse potential profile caused by edge doping. Note, however, that the measured edge response is weaker than in the simulations. It is likely that the bulk is more conducting in the real device due to disorder, which is nevertheless not strong enough to totally mask edge transport. The precise microscopic origin of the enhanced edge potential is also unclear, but may arise from the trapped molecules (e.g., water) between the bottom hBN and the SiO₂ substrate, consistent with the observed hole doping in SGS. We also note that the finer conductance oscillations in the simulated SGS [Figure 4d] are not reproduced in the measured SGS [Figure 2f]. It is likely they result from quantum interference or localization within narrow conducting strips near the edges. Whether such localization would be observed experimentally might strongly depend on the detailed shape of the edge potential, the bulk disorder level, and thermal broadening relative to the spacing of the energy levels.
In summary, we have used SGM to study subgap transport near the CNP of gapped moiré MLG devices. In one device, we observe a transition from bulk to edge response as the Fermi level approaches the CNP, suggesting currents flow at the edges. Guided by numerical simulations, we showed that transport can be explained by hole doping near the edges. The absence of such edge conduction in a much cleaner device also suggests a more complicated picture than the bulk-shunting edge states. These observations may serve as a starting point for improving the insulating behavior using local gates to compensate doping at the edges and motivate a study to determine how disorder, the gap size, and subgap transport are statistically correlated in these devices. Indeed, very recently, similar gapped Dirac system with lower charge disorder realized by dual gated bilayer graphene and graphite backgate has also shown much better insulating behavior near the Dirac point.

Direct SGM imaging of bulk and edge currents implicated in the VHE and at the SDPs would also be a natural extension of this work. Moreover, our results show how SGM could be used to examine transport in a wide range of low-dimensional materials, such as topological insulators and superconductor–semiconductor hybrids, where differentiating between trivial and topological edge modes could prove important for proposed applications in quantum computing and low power electronics.

**Associated Content**

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Additional information and figures (PDF)

**Author Information**

**Corresponding Authors**

*E-mail: zd235@cam.ac.uk.*

*E-mail: mrc61@cam.ac.uk.*

**ORCID**

Ziwei Dou: 0000-0001-5591-2750

Kenji Watanabe: 0000-0003-3701-8119

**Present Address**

V. (M.R.C.) Niels Bohr Institute, University of Copenhagen, Universitetsparken 5, 2100 København Ø.

**Author Contributions**

Z.D. conducted the low-temperature transport measurement and all the low-temperature scanning gate and spectroscopy measurement; S. Morikawa and S. Masubuchi prepared the stamped van der Waals heterostructure and fabricated the devices with supervision from T.M. and prepared them for the microscope with input from S.W.W.; Z.D. and M.R.C. conceived the idea, designed the experiments, analyzed the data and wrote the manuscript with useful discussions from C.G.S., A.C., S. Morikawa, S. Masubuchi, and T.M. A.C. performed the tight-binding simulations; C.M and O.K. performed and supervised the Raman spectroscopy. K.W. and T.T. provided the h-BN material; M.R.C. and T.M. supervised the research. All authors have given approval to the final version of the manuscript.

**Notes**

The authors declare no competing financial interest.

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