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Imaging Bulk and Edge Transport near the Dirac Point in Graphene Moiré Superlattices

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ABSTRACT: Van der Waals structures formed by aligning monolayer graphene with insulating layers of hexagonal boron nitride exhibit a moiré superlattice that is expected to break sublattice symmetry. Despite an energy gap of several tens of millielectronvolts opening in the Dirac spectrum, electrical resistivity remains lower than expected at low temperature and varies between devices. While subgap states are likely to play a role in this behavior, their precise nature is unclear. We present a scanning gate microscopy study of moiré superlattice devices with comparable activation energy but with different charge disorder levels. In the device with higher charge impurity (\(\sim 10^{10}\, \text{cm}^{-2}\)) and lower resistivity (\(\sim 10\, \text{k}\Omega\)) at the Dirac point we observe current flow along the graphene edges. Combined with simulations, our measurements suggest that enhanced edge doping is responsible for this effect. In addition, a device with low charge impurity (\(\sim 10^{9}\, \text{cm}^{-2}\)) and higher resistivity (\(\sim 100\, \text{k}\Omega\)) shows subgap states in the bulk, consistent with the absence of shunting by edge currents.

KEYWORDS: Graphene moiré superlattice, gapped Dirac Fermion system, scanning gate microscopy, topological edge states, valley Hall effects

The mobility of charge carriers in two-dimensional (2D) crystals of graphene is enhanced by encapsulating them between atomically flat layers of hexagonal boron nitride (hBN). Alongside improvements in device quality has come a rich array of physics arising from the van der Waals interaction between the 2D layers. Moiré superlattices, for instance, are formed by aligning the graphene lattice within a few degrees of the hBN [Figure 1a]. Strain-relaxation instance, are formed by aligning the graphene lattice within a few degrees of the hBN [Figure 1a]. Strain-relaxation and opens a gap in the quasiparticle density of states at the Dirac point. An energy gap of the order of 10 meV is typically extracted from thermally activated transport near the charge neutrality point (CNP) and shows the expected functional dependence on the moiré wavelength \(\lambda\). Activated transport is observed in the high temperature regime (>\(\sim 50K\)), but at lower temperature the resistivity at the CNP shows a much weaker temperature dependence. Similar behavior was observed previously in gapped bilayer graphene (BLG) and was attributed to the transition from thermal excitation over a bandgap to hopping conduction in the bulk via low energy states induced by charge disorder. Transport via edge states could also shunt the insulating bulk and lead to resistance saturation but no clear signatures of edge transport near the CNP were observed in conventional low-mobility graphene on SiO2 substrates. Edge states are, however, more likely to play a role in high-mobility graphene where bulk charge disorder is reduced. The distinction between edge and bulk transport in gapped graphene may also influence topological valley Hall effects (VHE), both in gapped moiré MLG and in the dual-gated BLG devices where nonzero Berry curvature is present due to broken sublattice symmetry. More recently, topological valley Hall effects (VHE) were observed in gapped moiré superlattices, where subgap states were found in the bulk due to the transition from thermal to hopping conduction.
symmetry. Understanding the current distribution in gapped graphene devices near the CNP is therefore a high concern. Indeed, recent Josephson interferometry in very short (∼100 nm) graphene junctions between superconducting Nb contacts shows a transition from bulk-to-edge transport occurs as the energy approaches the CNP in both gapped BLG and MLG.

In this work, we use scanning gate microscopy (SGM) to examine micron-sized MLG moiré superlattices. In particular, we focus on the current distribution near the main DP, where the gap due to moiré-induced broken inversion-symmetry and the mechanism for finite subgap conduction at low temperature is still not well understood. We present data taken from two devices with comparable activation energies ($E_a/2$ ∼ 14 meV) but different levels of disorder-induced energy broadening, $E_d$. In device D1, $E_d > E_a/2$, and the resistivity at the CNP saturates at ∼10 kΩ, comparable to the $h/e^2$, whereas in device D2, $E_d < E_a/2$, and the CNP resistivity (∼100 kΩ) is much higher than the $h/e^2$. Our SGM images show a response at the edges in D1 near the CNP. Despite its lower disorder, the SGM response in D2 is in the bulk. Using tight-binding simulations as a guide, we are able to explain our results by enhanced doping at the edges of D1. Our results therefore suggest that whereas edge transport plays an important role in shunting the gapped bulk in MLG moiré superlattice devices, it probably originates from external factors such as disorder and inhomogeneous doping rather than universal properties of gapped Dirac spectrum.

Our MLG encapsulated moiré superlattice devices [Figure 1a] are fabricated by mechanical exfoliation and the “pick-up” transfer technique of atomic layers. The MLG is sandwiched between a 50 nm top hBN layer and a 70 nm bottom layer hBN, and the whole structure is placed on a ∼290 nm SiO₂ layer formed on a doped Si substrate used for applying a back-gate voltage ($V_{BG}$). The devices are then patterned into Hall bars and Cr/Pd/Au ohmic contacts are made via one-dimensional edge contacts. An optical image of the device and circuit schematic is shown in Figure 1a. We use standard low-frequency lock-in techniques to determine the four-terminal resistance $R_{4T} = V_{xx}/I_{SD}$. For D1, we first confirm the presence of a moiré superlattice by observing secondary Dirac points (SDPs) in $R_{4T}(n)$, a well-known signature resulting from the modified bandstructure of moiré MLG$^3$ [Figure 1b]. The carrier density $n$ is calculated by $C_{BG}(V_{BG} - V_{CNP})$ where $C_{BG} = 1.124 \times 10^{-4}$ F/m² is the capacitance per area to the backgate extracted from quantum Hall measurements, and $V_{CNP} = 1.25$ V is the back-gate voltage at the charge neutrality point. From the SDP carrier densities we estimate $\lambda_M$ of ∼11 nm.$^3$ Figure 1c shows the resistivity $\rho$ (calculated from $R_{4T}$ ∼ $\rho L/W$, where $L = 11.5$ μm and $W = 1.3$ μm are the length and width, respectively) measured at $T = 4$ K as a function of $n$. From the full width at half-maximum (fwhm) of the Dirac peak (∼4 × 10−10 cm−2), we estimate disorder-induced broadening of ∼24 meV.$^3$ Figure 1d shows the temperature dependence of $\rho$ in the range of 4–200 K. The maximum resistivity at the CNP ($\rho_{CNP}$) is extracted from back-gate sweeps performed at each temperature and plotted against inverse temperature, resulting in an Arrhenius plot. We identify the following three regimes. (I) at high temperature (>70 K) the data are well described by a simple model for thermally activated transport across an energy gap $E_g$, $\rho_{CNP} \sim \exp\left(\frac{E_g}{2kT}\right)$, where $k$ is the Boltzmann constant. By fitting the data to this expression (red dashed line), we extract $E_g/2 = 160$ K (∼14 meV), consistent with previous work$^3$ and the gap expected from the estimated $\lambda_M$. (II) At intermediate temperatures (gray hatched lines), the weak temperature dependence is consistent with variable range hopping, and by fitting the resistivity to $\rho \sim \exp\left((T_0/T)^{1/2}\right)$, we deduce $T_0 \sim 2$ K. ($^3$) At low temperature, the resistivity saturates at ∼10 kΩ, comparable to the resistance quantum $h/e^2$ and consistent with earlier works.$^3$ In an effort to resolve the microscopic mechanism behind the flattening of $\rho_{CNP}(T)$ in regime (III), we use SGM at $T \sim 4$ K. SGM is a well-established technique for visualizing local variations in the electronic properties of 2DESes and involves monitoring the conductance of a mesoscopic device.

Figure 1. Transport characterization of MLG moiré Hall bar device (D1). (a) Left: moiré superlattice formed by aligning monolayer graphene (black) with hBN substrate (magenta). The red hexagon depicts a moiré unit cell. Right: optical image of the device with a schematic of the measurement setup (scale bar 1 μm). (b) Four-terminal longitudinal resistance $R_{4T} = V_{xx}/I_{SD}$ as a function of $n$ showing secondary Dirac points ($T = 4$ K). (c) Resistivity as a function of carrier density ($T = 4$ K). (d) Arrhenius plot of $\rho$ at the CNP as a function of $1/T$. Three regions are seen: (I) exponential dependence of $\rho$ on $1/T$ at high temperature; (II) variable-range hopping at immediate temperature (hatched gray); (III) constant $\rho$ at low temperature.
while scanning a sharp metallic tip over its surface. Figure 2a illustrates the SGM setup used in our experiments, where a constant current ($I_{SD} = 100$ nA) is driven between the two ends of the superlattice while measuring the voltage $V_{xx}$. The tip is lifted $\sim 100$ nm above the top hBN layer and locally gates the graphene. The scanned area in (b,c) are marked by the red dashed rectangle (tip not to scale, scale bar: 1 $\mu$m). SGM image of $R_{4T} = V_{xx}/I_{SD}$ as a function of tip position when globally (b) $n \sim -2.5 \times 10^{10}$ cm$^{-2}$ and (c) $n \sim 0$ cm$^{-2}$. Edges of the device are outlined by dotted lines and each grid represents 1 $\mu$m. Blue dashed oval depicts the region to the long-range tip gating effect. Insets show $R_{4T}(x)$ averaged over all $x$. (d) Sequence of higher-resolution SGM images of $R_{4T}$ in the area marked by the black rectangle in (b). The images are taken at different $n$: $-2.11 \times 10^{10}$ cm$^{-2}$ (blue); $-1.05 \times 10^{10}$ cm$^{-2}$ to $0$ with $-0.21 \times 10^{10}$ cm$^{-2}$ intervals (green to magenta); $0.14 \times 10^{10}$ to $0.70 \times 10^{10}$ cm$^{-2}$ with $-0.14 \times 10^{10}$ cm$^{-2}$ intervals (magenta to gray). The device edges are outlined with dotted lines; each grid represents 0.5 $\mu$m. Evolution of a “hotspots” is marked by blue dashed circle. (e) SGS as a function of $n$ along the black arrow in (d). The SGS is plotted in five panels with different color scales for clarity. Corresponding values of $n$ for the images in (d) are marked with colored circles. (f) Same SGS as (e) but with $V_T^{*} \sim +1.5$ V. (g) Reference $R_{4T}(n)$ without the tip. Plots in panels e–g share the same horizontal axis. The CNP is marked by the vertical dashed line.

Figure 2. Scanning gate microscopy of device D1 ($T = 4$ K). (a) Optical image of the device and measurement setup used for SGM. A dc biased AFM tip ($V_T^{*} \sim +0.5$ V, see main text) is lifted $\sim 100$ nm above the top hBN layer and locally gates the graphene. The scanned area in (b,c) are marked by the red dashed rectangle (tip not to scale, scale bar: 1 $\mu$m). SGM image of $R_{4T} = V_{xx}/I_{SD}$ as a function of tip position when globally (b) $n \sim -2.5 \times 10^{10}$ cm$^{-2}$ and (c) $n \sim 0$ cm$^{-2}$. Edges of the device are outlined by dotted lines and each grid represents 1 $\mu$m. Blue dashed oval depicts the region to the long-range tip gating effect. Insets show $R_{4T}(x)$ averaged over all $x$. (d) Sequence of higher-resolution SGM images of $R_{4T}$ in the area marked by the black rectangle in (b). The images are taken at different $n$: $-2.11 \times 10^{10}$ cm$^{-2}$ (blue); $-1.05 \times 10^{10}$ cm$^{-2}$ to $0$ with $-0.21 \times 10^{10}$ cm$^{-2}$ intervals (green to magenta); $0.14 \times 10^{10}$ to $0.70 \times 10^{10}$ cm$^{-2}$ with $-0.14 \times 10^{10}$ cm$^{-2}$ intervals (magenta to gray). The device edges are outlined with dotted lines; each grid represents 0.5 $\mu$m. Evolution of a “hotspots” is marked by blue dashed circle. (e) SGS as a function of $n$ along the black arrow in (d). The SGS is plotted in five panels with different color scales for clarity. Corresponding values of $n$ for the images in (d) are marked with colored circles. (f) Same SGS as (e) but with $V_T^{*} \sim +1.5$ V. (g) Reference $R_{4T}(n)$ without the tip. Plots in panels e–g share the same horizontal axis. The CNP is marked by the vertical dashed line.
the effective tip potential $V_T^* = V_T - V_0 = -0.5$ V, reducing the resistance of the hole-doped channel and producing the feature within the oval dashed line. When the tip is over the channel, $V_0$ is measured to be $-1$ V, and $V_T^* = 0.5$ V, giving rise to the observed resistance enhancement (see Supporting Information, Section VI). With the global doping level at the CNP [Figure 2c, bottom image], a $\sim 1$ kΩ enhancement in $R_{IT}$ is seen only when the tip is over the edges of the flake [see plot of averaged $R_{IT}$ (y) in Figure 2c], which cannot be explained by a spatially dependent contact potential (see Supporting Information, Section VI).

We investigate more closely by capturing a sequence of high-resolution scans at different carrier densities over the range from $-2.1 \times 10^{10}$ cm$^{-2}$ to $0.7 \times 10^{10}$ cm$^{-2}$ (note the unequal carrier density intervals denoted in the caption) and plot the SGM micrographs in Figures 2d. The scanned region is outlined in Figure 2b by the black rectangular box and the device edges are marked as black dotted lines. The peaks of the Lorentzian trajectories associated with the sites in (e) are marked by blue dashed circle. (f) SGS as a function of n and distance y along the vertical white arrow line drawn in (e). Corresponding values of n for the images in (e) are marked with colored circles. (g) Reference $R_{IT}(n)$ without the tip. (h) SGS as a function of n and distance x along the horizontal white arrow in (e). The "hotspots" (defined here as an isolated region of tip position where the resistance is higher than the background) that are initially over the bulk channel split laterally and migrate toward the edges. At the CNP, some hotspots fragment further along the edge (see the dashed circles). To track the bulk to edge transition we perform scanning gate spectroscopy (SGS), which involves stepping the tip position along a line and sweeping the global carrier density at each point. Plotting the resulting $R_{IT}(n, y)$ allows us to determine the evolution of a particular feature with both high energy and high spatial resolution in one dimension.

Figure 3. Scanning gate microscopy of device D2 ($T = 4$ K). (a) Optical image of the device and measurement setup used for SGM (tip not to scale, scale bar: 1 μm). Scanned area is marked by the red dashed rectangle. (b) Comparison between the resistivity $\rho(n)$ ($T = 4$ K) of D2 (solid) and D1 (dashed), showing a narrower fwhm and $\sim 10$ times higher resistivity at the charge neutrality point for D2. (c) Arrhenius plot of $\rho$ at the CNP as a function of $1/T$. Three regions similar to D1 (Figure 1) can be identified. Dashed line marks the low-temperature resistivity of D1. (d) $R_{IT}(n)$ plotted over a wider range of n to show the SDPs for D2 (solid) and D1 (dashed) ($T = 4$ K). (e) Sequence of SGM images of $R_{IT} = V_{SD}/I_{SD}$ captured at equal n intervals between $-2.5 \times 10^{9}$ and $7.4 \times 10^{9}$ cm$^{-2}$. Localized sites of higher resistance are marked by the blue dashed circles [see (h) for corresponding spectroscopy]. The device boundaries are marked with dotted lines; each grid represents 1 μm. (f) $R_{IT}(n)$ as a function of n and distance y along the vertical white arrow line drawn in (e). Corresponding values of n for the images in (e) are marked with colored circles. (g) Reference $R_{IT}(n)$ without the tip. (h) SGS as a function of n and distance y along the horizontal white arrow in (e). The peaks of the Lorentzian trajectories associated with the sites in (e) are marked by blue dashed circle. (i) SGS as a function of n and distance x along the horizontal white arrow in (e). The peaks of the Lorentzian trajectories associated with the sites in (e) are marked by blue dashed circle. (j) SGS as a function of n and distance x along the horizontal white arrow in (e). The peaks of the Lorentzian trajectories associated with the sites in (e) are marked by blue dashed circle.
altogether with more positive tip bias as the device becomes electron doped, such features disappear. A feature resembling the transition back to bulk response is seen predominantly occurs for hole doping. While in Figure 2e, the charge carrier density disorder of D2 is ~0.4 × 10^{10} cm^{-2} (or ~7 meV), a factor of 3 lower than D1 (plotted in reference as the dashed line) and is comparable with the best quality reported in literature.\textsuperscript{3,4,6} The temperature dependence [Figure 3c] shows an activated region with similar transport gap ~28 meV and plateau at low temperature. Figure 3d confirms the moiré superlattice of the sample with SDPs translated to similar moiré wavelength ~11 nm. Therefore, D2 has the same gap size as D1 but with lower bulk disorder. From earlier proposals\textsuperscript{10,12,18} edge transport is expected to be more pronounced in D2. To test this, we perform the equivalent SGM and SGS using two-terminal measurements [Figure 3a]. Figure 3f shows a series of SGM images of \( R_{2T} = V_{SD}/I_{SD} \) with constant voltage \( V_{SD} = 1 \text{ mV} \) taken with the same scan conditions as Figure 2 at different values of \( n \) around the CNP. We do not observe an edge response but only a few regions in the bulk where the tip can affect transport [seen the dashed circles in Figure 3e]. We follow the same method employed for D1 and perform SGS along the \( y \)-direction perpendicular to the edge. The spectroscopy plotted in Figure 3f shows a weak gating effect that produces an overall shift in the back-gate voltage of the CNP but with no increase in \( R_{2T} \) at the edge near the CNP \( [R_{2T}(n)] \) without the tip is plotted in Figure 3g for...
reference. The strong ~ MΩ variations due to the tip (as large as ~50% of the total R_{CT}) are clearly concentrated in the bulk.

We perform SGS by moving the tip along a line, this time parallel to the channel [arrow along x in Figure 3e] and intersecting the sites marked by the dashed circles. The spectroscopy is plotted in Figure 3h. When the tip is in the vicinity of each site, a peak in R_{CT}(n) appears at higher n. This peak splits from the main Dirac point and follows a Lorentzian-shaped trajectory as a function of n. To emphasize these trajectories we take the derivative dR_{CT}/dn in the area marked by the white box in Figure 3h, and plot the result in Figure 3i. The trajectories can be fitted by Lorentzians with fwhm of ~0.5 μm and origins separated by Δx ~ 1 μm. Line sections [Figure 3i] show that height of each peak (P_{x=}) is comparable to the amount by which the main Dirac peak (P_{+}) is reduced (~1 MΩ), suggesting the sites add in series to yield the total resistance at the DP. We confirmed that sites are not associated with inhomogeneous strain in the sample28 by checking the 2D peak width using scanning Raman spectroscopy (see Supporting Information Section IX). Despite its lower bulk disorder, D2 does not show pronounced edge conduction2,10,12 and the transport appears mediated via subgap states in the bulk. The absence of edge shunting is also noteworthy, given that the resistivity at CNP plateaus are higher than h/e^2.12 One natural explanation is that transport is via a series of electron–hole puddles forming a bulk percolation path.12 High quality devices typically have ~100 nm size puddles29,30 consistent with a small number of conducting sites separated by insulating regions.

Several mechanisms could give rise to the observed edge transport in D1. Possible scenarios include zigzag edge states without11 or with energy dispersion under more sophisticated Hamiltonians31 and states with enhanced localization length due to nontrivial bulk band topology from the broken sublattice symmetry.10,12 However, because these states are not topologically protected from intervalley scattering, they localize easily and do not participate in transport.10 One of the main reasons for this is the presence of edge shunting, which is also evident in the experimental bulk. The SGS [Figure 4d] is plotted in five panels with different color scales similar to Figure 2. The tip passes over the two edges of the channel at y = 0 and 1 μm (x_f = 1 μm) and the energy gap (±7.5 meV) is marked by the dashed vertical lines. This plot captures some key features of the experimental SGS in Figure 2, namely a clear enhancement of the resistance appears at the edges at around −7.5 meV in the hole-doped regime and diminishes as E_T is tuned away from CNP.

To illustrate how the above SGM features relate to the enhanced doping at the edges, in Figure 4e,f we plot how the underlying current distributions j(x,y) in Figure 4b,c change in the presence of the tip, Δj = j(x,y) − j_0(x,y). At the edge of the flake, the bandgap (the region between the red and blue solid lines) is lifted to higher energy [Figure 4e, right]. Hence, when E_T = −12.5 meV [blue circle in Figure 4d], the edge has a higher hole density than the bulk. Although the current is enhanced along the edges, the effect of the negative tip potential energy is weak because the bulk is conductive. The simulated Δj map [Figure 4e] consequently displays weak ambipolar modulations of the current and there is no enhancement at the edges. By contrast, when E_T = 0, the Δj map [Figure 4f] shows a strong response along the edge because the conduction and valence bands are lowered and the hole density reduces [see band edge profile, Figure 4f]. This gaps out the current flow at the edges and increases the resistance. In the inset of Figure 4f, the same Δj map is plotted with positive tip potential. In this case, the edge current is locally enhanced and only weakly affects the global resistance, consistent with the absence of an SGS response. We experimentally observe an edge response only with positively biased tip (or negative tip potential energy), while no edge response is seen with negatively biased tip. This is further consistent with enhanced p-type doping (see Supporting Information Section VII). Within this framework potential disorder along the channel creates the hotspots in Figure 2, while their migration toward the edge reflects the transverse potential profile caused by edge doping. Note, however, that the measured edge response is weaker than in the simulations. It is likely that the bulk is more conducting in the real device due to disorder, which is nevertheless not strong enough to totally mask edge transport.13 The precise microscopic origin of the enhanced edge potential is also unclear, but may arise from the trapped molecules (e.g., water) between the bottom hBN and the SiO2 substrate, consistent with the observed hole doping in SGS.35 We also note that the finer conductance oscillations in the simulated SGS [Figure 4d] are not reproduced in the measured SGS [Figure 2f]. It is likely they result from quantum interference or localization within narrow conducting strips near the edges.27 Whether such localization would be observed experimentally might strongly depend on the detailed shape of the edge potential, the bulk disorder level, and thermal broadening relative to the spacing of the energy levels.
In summary, we have used SGM to study subgap transport near the CNP of gapped moiré MLG devices. In one device, we observe a transition from bulk to edge response as the Fermi level approaches the CNP, suggesting currents flow at the edges. Guided by numerical simulations, we showed that transport can be explained by hole doping near the edges. The absence of such edge conduction in a much cleaner device also suggests a more complicated picture than the bulk-shunting edge states.  

Direct SGM imaging of bulk and edge currents implicated in the VHE and at the SDPs would also be a natural extension. Moreover, our results show how SGM could be statistically correlated in these devices. Indeed, very recently, dual gated bilayer graphene and graphite backgate has also been performed and supervised the Raman spectroscopy. K.W. and C.G.S. conceived the idea, designed the experiments, analyzed the measurements; S.W.W.; Z.D. and M.R.C. performed the tight-binding simulations; C.M and O.K. performed and supervised the Raman spectroscopy. K.W. and T.T. provided the h-BN material; M.R.C. and T.M. supervised the research. All authors have given approval to the final version of the manuscript.

Notes
The authors declare no competing financial interest.

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